

## MULTIPLE ANNEAL INDUCED DISORDERING

The present invention relates to quantum well intermixing (QWI) techniques suitable for modifying an energy bandgap during the formation of optical semiconductor devices. In particular, the invention relates to QWI techniques in which spatial control of the QWI process can be effected so as to achieve differing bandgap shifts across a wafer, device or substrate surface.

10 A vast body of research exists in the field of QWI. The QWI process consists in the selective disordering of the composition of the thin layers that form quantum wells, which results in a change of energy levels within each well causing the energy bandgap to shift. This allows one to alter the emission and absorption wavelengths of the intermixed material.

15 A variety of QWI techniques have been developed including: impurity-induced, impurity free (dielectric cap), implantation-induced and laser induced methods. QWI has been demonstrated in a range of material systems, including GaAs/AlGaAs and InP/AlInGaAs(P).

20 Much effort recorded in the prior art (prior art references are given in the Annex to this description, as referred to in square parentheses) has been directed to achieving a dual-bandgap process, where the emphasis is on obtaining a large differential shift between areas of reduced shift (nominally the as-grown bandgap) and intermixed areas. Various techniques have been proposed to enhance control over bandgap shifts, e.g. varying the material [1, 2], deposition conditions [3, 4], stoichiometry [5], size [6] and thickness [7-9] of the dielectric cap in impurity-free processes; ion irradiation dose [8, 10], laser exposure [11-14], surface coverage/resolution effects [15], and, 30 most commonly, anneal temperature and duration in almost all of the above

reports. Not all of these approaches, however, can be used to create multiple, i.e. greater than 2 bandgaps on a single wafer – by temperature adjustment alone one cannot obtain more than one shift.

5 Most generally, multiple bandgaps can be created using a core dual-bandgap process with one of the following approaches:

1. Repeated [10, 15-21] / variable dose [12-14] exposure-anneal combinations;
2. Choice of dielectric caps of different material [2-5] and interface  
10 effects [1, 22-24];
3. QWI barrier masks [7, 9, 25, 26] and caps of varying thickness [7, 8];  
and
4. Spatial / resolution effects [6, 15]

15 Despite the abundance of QWI techniques, there is a scarcity of prior art where these techniques could be used in a controlled manner to define multiple bandgaps on a common substrate.

A first prior art approach is based upon use of repetitive cycles of ion-  
20 implantation / plasma exposure and rapid thermal anneal (RTA) at high-temperature to obtain required bandgap shifts [10, 15-21]. This approach has been used to tune the wavelength of quantum-well lasers [17, 27] and infrared photodetectors [20, 21]. Repetitive steps are employed to achieve larger cumulative shifts than those attainable with a single-anneal process  
25 [17-20, 28].

To achieve multiple bandgaps in selected areas using the above approach, one needs to pattern these areas and expose them to the implantation / anneal cycle, repeating the procedure for each shifted bandgap required, as  
30 suggested in [20]. Here, the underlying assumption is that subsequent

intermixing cycles have no effect on the shifts in the areas processed at earlier stages as long as they are protected during subsequent exposures. This assumption is believed to be true for a range of ion-implantation techniques, as the shifts are best controlled by adjusting the irradiation dose  
5 [18, 20].

The situation is more complicated where anneal conditions are used to control the amount of intermixing. RTA affects areas of all bandgaps regardless of the order in which they were exposed, thus possibly shifting  
10 the bandgaps created previously even further. In the case of a combined implantation/anneal cycle, one can circumvent this problem by carrying out the multiple patterning and irradiation steps first, followed by a common anneal step at the end.

15 In report [29], multiple bandgaps were created using multiple anneals with a sacrificial ion implant layer, which was selectively removed in areas requiring no further shifting during subsequent anneals. In locations where the sacrificial layer was removed, QWI suppression was achieved such that subsequent anneals caused substantially no further bandgap shift in those  
20 locations.

Such a solution is not possible, however, in QWI RTA-enabled processes where shifts are induced through impurity diffusion and/or by dielectric caps, in particular, in a sputtering-induced disordering (SID) process. The  
25 SID process involves sputter deposition of an impurity (such as sulphur, zinc, silicon, fluorine, copper, germanium, tin, selenium, etc onto the material surface) followed by a high-temperature anneal. Suppression of QWI (zero bandgap shift) is achieved by protecting the respective other areas of the substrate with a layer of PECVD-deposited silica.

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Here, the high-temperature-induced creation and interdiffusion of defects during the anneal is the prime intermixing mechanism, which cannot always be fully suppressed by the removal of the defect / impurity source in subsequent anneal stages. In other words, while in irradiation / exposure-based processes the action of the intermixing agent (e.g. implantation dose or the defect-rich layer of [29]) can be limited to a particular intermixing step, in cap-based processes such action cannot always be suppressed in subsequent intermixing steps even by the removal of the QWI cap. Therefore, in the latter case, each subsequent RTA stage will uncontrollably affect the shifts obtained in all the previous steps.

In a second prior art technique, [15, 16] and many others, it is proposed to vary anneal duration to control QWI shift using dielectric caps. However, the only mention of a multiple anneal process being used in conjunction with a dielectric cap is found in [16]. InGaAs/InAlAs multiple-quantum well (MQW) structures were partially disordered by the deposition of a  $\text{Si}_3\text{N}_4$  dielectric cap followed by repeated RTAs at 850 degrees C for 1 to 5 seconds. The only reason given for doing so is to achieve a larger cumulative shift (43 nm) than that attainable with a single-anneal process. The paper [16] does not consider the creation of multiple bandgaps on a single substrate, nor the difficulty in overcoming unwanted further shifts in previously intermixed regions.

It is an object of the present invention to provide a QWI process that is capable of providing multiple bandgap shifts on a single device substrate using impurity diffusion and/or dielectric cap-based QWI.

According to one aspect, the present invention provides a method for producing multiple quantum well intermixed (QWI) regions having different bandgaps on a single substrate, comprising the steps of:

- a) patterning the surface of the substrate with QWI-initiating material in first regions of the surface;
- b) conducting a first thermal processing cycle on the substrate to generate a first bandgap shift in the first regions;
- 5 c) patterning the surface of the substrate with QWI-initiating material in second regions of the surface, distinct from said first regions; and
- d) conducting a second thermal processing cycle on the substrate to generate a second bandgap shift in the second regions, and to generate a cumulative bandgap shift in the first regions, the cumulative bandgap shift  
10 being the cumulative result of said first and second thermal processing cycles.

According to another aspect, the present invention provides a method for determining required parameters for each of the thermal processing cycles of  
15 the method defined immediately above, comprising the steps of:

determining whether the process for generating cumulative bandgap shifts resulting from successive thermal processing cycles is symmetric or asymmetric;

if the process is symmetric, then determining the thermal process  
20 conditions required for each one of a plurality of cumulative bandgap shifts  $BG_1$  to  $BG_N$  by successive use of at least one sample through a thermal process sequence  $A_N$  to  $A_1$ , where  $A_1$  is the thermal process required to obtain  $BG_N$  from  $BG_{N-1}$ ;  $A_2$  is the thermal process required to obtain  $BG_{N-1}$  from  $BG_{N-2}$ ; etc.; through to  $A_N$  being the thermal process required to obtain  
25  $BG_1$  from  $BG_0$ ; and

if the process is asymmetric, then determining the thermal process conditions required for each one of the plurality of cumulative bandgap shifts  $BG_1$  to  $BG_N$  by use of a plurality of samples through a partial or complete thermal process sequence in the order  $A_1$  to  $A_N$  for each one of the  
30 bandgap shifts required.

Embodiments of the present invention will now be described by way of example and with reference to the accompanying drawings in which:

Figure 1 is a graph illustrating the effects of a subsequent thermal  
5 anneal process on a substrate after a QWI-initiating layer has been stripped;

Figure 2 is a graph illustrating impurity concentration as a function of depth through the substrate before and after QWI processing;

Figure 3 is a schematic diagram of the device substrate during various stages of the QWI processing steps according to one embodiment of the  
10 present invention;

Figure 4 shows schematically initial and subsequent bandgap shifts, represented by photoluminescence wavelength shift, effected in different regions of the substrate during the processing steps as applied to the structures of figure 3;

Figure 5 shows experimentally measured initial and subsequent  
15 bandgap shifts, represented by photoluminescence wavelength shift, effected in different regions of the substrate during the processing steps as applied to the structures of figure 3, in 3D graph form (figure 5a) and in 2D graph form (figure 5b);

Figure 6 shows bandgap shifts, represented by photoluminescence  
20 wavelength shift, as a function of anneal time and anneal temperature in 3D graph form (figure 6a) and in 2D graph form (figure 6b); and

Figure 7 shows a procedural flow chart for generating the required anneal conditions for a desired set of bandgaps.

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The process of the present invention allows multiple bandgaps to be defined in a controlled manner, in several intermixing steps, on the same wafer or substrate. The process is compatible, *inter alia*, with Al-quaternary InP material on a semi-insulating (SI) InP substrate, and thus can be used to  
30 fabricate high-frequency optoelectronic devices. This enables a plethora of

component integration possibilities whereby active and passive components, each bandgap-tuned as required, can be fabricated on a common substrate as part of a photonic integrated circuit and/or a single integrated device. Examples include semiconductor optical amplifier (SOA)-preamplified  
5 modulators, photodetectors and switches, extended-cavity lasers, wavelength-detuned laser arrays, demultiplexors, etc.

It has been established that in many QWI RTA-enabled processes where shifts are induced through impurity diffusion and/or by QWI-initiating  
10 dielectric caps, after initial annealing, the removal of the QWI-initiating dielectric cap after one thermal processing step has little effect on the shifts obtained in subsequent thermal processing steps. In other words, following the first QWI step, the shifts obtained with and without the QWI-initiating cap on are almost identical. This effect is clearly illustrated in figure 1.

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Figure 1 shows the photoluminescence wavelength shift resulting from various QWI processes, being a direct measurement of the bandgap shift. In the reference sample indicated by line 10, no QWI-initiating cap is applied to the substrate. In the first sample, indicated by dotted line 11, a QWI-  
20 initiating cap is applied for the anneal process step 1, and stripped for the subsequent anneal process step 2. By way of contrast, in the second sample, indicated by dashed line 12, a QWI-initiating cap is applied for the anneal process step 1 and maintained for the subsequent anneal process step 2. It can readily be seen by comparison of lines 11 and 12 that the stripping of the  
25 QWI-initiating cap after the first anneal has only a marginal effect on the shift induced by the second anneal. The anneal process 1 and anneal process 2 each correspond to an isothermal anneal for 2 minutes at 615 °C on a multiple quantum well structure in InP / InAlGaAs material.

Further processing, such as removal of the top semiconductor layers in contact with the QWI-initiating layer, will not normally enable the total suppression of further bandgap shifting with subsequent thermal processing. Figure 2 shows the concentration of the QWI-initiating impurity (in this example, copper) as a function of depth into the substrate (indicated by etch time in seconds) for both unannealed samples (profile 20) and for samples annealed at 750 degrees C for 3 minutes (profile 21), as measured by SIMS. The QWI-initiating cap layer 22, the impurity-rich InGaAs layer 23, quantum well layers 24 and mechanically supporting substrate 25 are shown to the right of the graph. It is noted that a number of possible impurity materials may be used for QWI initiation. These include sulphur, zinc, silicon, fluorine, copper, germanium, tin and selenium.

For the avoidance of doubt, it is noted that throughout the present specification and particularly including the claims, for brevity and clarity the expression 'substrate' is used in a general sense to include the mechanically supporting and 'original' substrate 25 and all further material layers in existence above that original substrate at the time of a subsequent process step. In other words, the expression 'substrate' is intended to cover the totality of previously processed material and layers to which a process or further process (e.g. layer deposition or thermal treatment) is to be applied. The original 'raw material' substrate will be referred to as the mechanically supporting substrate, although it will be understood that this too may change its physical and chemical characteristics during processing.

As shown by the concentration distribution of the QWI-enabling impurity measured by the SIMS analysis before and after annealing with the QWI-initiating cap on, the impurity has penetrated the material very deeply, far beyond the active region containing the quantum wells. The high concentration of the impurity that has diffused into the material after the

initial anneal step can cause further intermixing at subsequent anneals even if the impurity source located in the top layers is removed. Therefore, suppression of further bandgap shifting by removal of the QWI-initiating layer 22 (or layers 23 below) is not likely to be successful in many processes.

The multiple anneal process of the present invention is illustrated for the multi-bandgap intermixing of InP/InGaAs material on a SI InP substrate using the dual-bandgap SID process described above.

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Figure 3 illustrates the process steps carried out on a device substrate for intermixing InP/AlInGaAs material to obtain four different bandgaps (including the as-grown one). The topmost semiconductor layer 30 of an as-grown epi wafer structure is represented in Figure 3-1. The target bandgaps are referred to herein, in order of increasing shift, as BG<sub>0</sub>, BG<sub>1</sub>, BG<sub>2</sub> and BG<sub>3</sub>. BG<sub>0</sub> corresponds to the as-grown (non-intermixed) bandgap.

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In step 2 (figure 3-2), the substrate 30 is covered with a blanket layer of PECVD silica 31.

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In step 3 (figure 3-3), a photoresist layer 32a is deposited onto the wafer and is photolithographically patterned to expose regions 33 targeted to have the largest eventual bandgap BG<sub>3</sub>.

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In step 4 (figure 3-4), the PECVD silica 31 in the exposed windows 33 is removed by wet-etching using an HF-based etchant.

In step 5 (figure 3-5), a QWI-initiating cap 34a consisting of a layer of an impurity and then a layer of silica is sputtered across the wafer.

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In step 6 (figure 3-6), the QWI-initiating cap 34a is lifted off in all areas where it overlies the photoresist layer 32a using a conventional photoresist lift off process to leave the QWI-initiating cap 34a in the BG<sub>3</sub> region 33 and to leave the QWI-inhibiting PECVD layer 31 in all regions where QWI is to be suppressed.

In step 7 (figure 3-7), the wafer is intermixed by high-temperature rapid thermal anneal, to generate a bandgap shifted region 33a (eventually targeted to have the largest bandgap shift BG<sub>3</sub>).

In step 8 (figure 3-8), a photoresist layer 32b is deposited onto the wafer 30 and is photolithographically patterned to expose regions 35 targeted to have the next largest bandgap BG<sub>2</sub>.

In step 9 (figure 3-9), the PECVD silica 31 in the exposed windows 35 is removed by wet-etching using an HF-based etchant.

In step 10 (figure 3-10), a QWI-initiating cap 34b consisting of a layer of an impurity and a layer of silica is sputtered across the wafer.

In step 11 (figure 3-11), the QWI-initiating cap 34b is lifted off in all areas where it overlies the photoresist layer 32b using a conventional photoresist lift off process to leave the QWI-initiating cap 34b in the BG<sub>2</sub> region 35 and to leave the QWI-inhibiting PECVD layer 31 in all regions where QWI is to be suppressed.

In step 12 (figure 3-12), the wafer is intermixed by high-temperature rapid thermal anneal, to generate a bandgap shifted region 36a (eventually targeted to have the next largest bandgap shift BG<sub>2</sub>) and to further modify the bandgap of the BG<sub>3</sub> region, 33b.

In step 13 (figure 3-13), a photoresist layer 32c is deposited onto the wafer 30 and is photolithographically patterned to expose regions 37 targeted to have the smallest shifted bandgap  $BG_1$ .

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In step 14 (figure 3-14), the PECVD silica 31 in the exposed windows 37 is removed by wet-etching using an HF-based etchant.

10 In step 15 (figure 3-15), a QWI-initiating cap 34c consisting of a layer of an impurity and a layer of silica is sputtered across the wafer.

15 In step 16 (figure 3-16), the QWI-initiating cap 34c is lifted off in all areas where it overlies the photoresist layer 32c using a conventional photoresist lift off process to leave the QWI-initiating cap 34c in the  $BG_1$  region 37 and to leave the QWI-inhibiting PECVD layer 31 in all regions where QWI is to be suppressed.

20 In step 17 (figure 3-17), the wafer is intermixed by high-temperature rapid thermal anneal, to generate a bandgap shifted region 38 (with bandgap shift  $BG_1$ ); to further modify the bandgap of the region 36b to its final bandgap shift  $BG_2$ , and to further modify the bandgap of the region 33c to its final bandgap shift  $BG_3$ .

25 In step 18 (figure 3-18), the QWI-initiating cap layers 34a, 34b and 34c, as well as any QWI-inhibiting cap layer 31 are removed by wet-etching using an HF-based etchant. The result is the regions 39 having bandgap  $BG_0$  (unshifted); regions 38 having bandgap shift  $BG_1$ ; regions 36b having bandgap shift  $BG_2$ ; and regions 33c having bandgap shift  $BG_3$ .

It will be understood that the process described above can be extended in principle to any number of different bandgaps.

It is important to note that the target bandgaps  $BG_0$  to  $BG_3$  are not fully defined until after the final anneal step (step 17 above). The evolution of the amount of intermixing in Regions 0 to 3 designated for bandgaps  $BG_0$  to  $BG_3$  with each anneal step is illustrated in figure 4. The as-grown case of zero shifts in all regions is shown in figure 4-1. With each anneal step represented in figures 4-2, 4-3 and 4-4, the shifts are accumulated in the regions subjected to multiple anneals with the QWI-initiating cap on and only meet the target shifts  $BG_0$  to  $BG_3$  on completion of the final (in this case, third) anneal step of figure 4-4.

In a general aspect, it will be recognised that the above process effectively comprises patterning the substrate surface with QWI-initiating material in first regions of the surface, conducting a first thermal process on the substrate to generate a first bandgap shift in the first regions, patterning the surface of the substrate with QWI-initiating material in second regions of the surface, distinct from said first regions, and conducting a second thermal process on the substrate to generate a second bandgap shift in the second regions, and to generate a cumulative bandgap shift in the first regions, the cumulative bandgap shift being the cumulative result of said first and second thermal processing cycles.

This can be followed by further steps of patterning the surface of the substrate with QWI-initiating material in third regions of the surface, distinct from said first regions and said second regions, and conducting a third thermal processing cycle on the substrate to: (i) generate a third bandgap shift in the third regions, (ii) generate a cumulative bandgap shift in the second regions, the cumulative bandgap shift in the second regions being the

cumulative result of the second and third thermal processing cycles; and (iii) generate a further cumulative bandgap shift in the first regions, the cumulative bandgap shift in the first regions being the cumulative result of the first, second and third thermal processing cycles.

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Thereafter, further steps can be added for still more bandgaps, patterning the surface of the substrate with QWI-initiating material in other regions of the surface, distinct from all regions of the surface previously covered with QWI-initiating material, and conducting a subsequent thermal processing cycle to generate a bandgap shift in the other regions, and to generate cumulative bandgap shifts in all regions previously covered with QWI-initiating material prior to the most recent patterning step, the cumulative bandgap shifts each being the cumulative result of all thermal processing cycles to which the respective region has been exposed since being first covered with the QWI-initiating material.

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The results of an experimental demonstration of the above process are given in figure 5, where target photoluminescence shifts of 20, 30 and 40 nm are obtained in InP/AlInGaAs MQW material using the core SID process after a series of three anneals each of identical duration (2 min) but at different temperatures (605, 610 and 646 °C). The intermediate shifts at each anneal stage are also shown.

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As pointed out earlier, QWI shifts can be controlled by a combination of anneal temperature and time. This is further demonstrated using the SID process, with the results presented in figure 6 as a 3D surface in the anneal temperature-time-bandgap shift space (figure 6a) and as a contour plot (figure 6b). The data are obtained by sequential isothermal anneals of one sample and measurement of photoluminescence shifts after each step, with the procedure repeated with different samples for a range of temperatures.

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The shifts obtained by a series of individual anneals and a single long anneal of the same total duration were found to be almost identical.

The process described above is capable of achieving the same range of bandgap shifts as are obtained by a conventional single anneal process that obtains just two bandgaps (i.e. the as-grown bandgap and the QWI bandgap). Furthermore, a large number of sequentially increasing bandgaps with arbitrary spacing are achievable. This is evident from the continuity of the data of figures 6a and 6b, which confirm that a series of anneals of the same duration and temperature can be replaced with a single anneal, of a different duration and temperature, that produces an identical shift (i.e., any point that lies on the same iso-shift contour line of figure 6b).

The same is true for a series of non-isothermal anneals of varying durations, provided that the total shift is less than the full shift obtainable with the dual-bandgap process (i.e., the saturation of bandgap shift as a function of anneal temperature / time). Note that this assumption is not applicable to techniques that utilise multiple anneals to achieve shifts that cannot be obtained with a single-stage anneal, as in [16], which would result in singularities in plots similar to those of figure 6. In that case, while the general concept described herein may still be used, only limited combinations of bandgaps may be achieved.

To define N shifted bandgaps,  $BG_1...BG_N$ , with  $BG_0$  being the zero-shifted one, the multiple anneal process requires N anneal steps  $A_1...A_N$ , with each step having unique anneal conditions (temperature and duration). The least-shifted bandgap,  $BG_1$ , is obtained by the final anneal step  $A_N$ , which we express as  $A_N(BG_0) = BG_1$ . The largest bandgap,  $BG_N$ , conversely, is the result of all the anneals  $A_1...A_N$ , which can be written as

$A_1 \rightarrow A_2 \rightarrow \dots \rightarrow A_N(BG_0) = BG_N$ .

For the simplest case of  $N = 2$ , having established  $A_2$  for  $A_2(BG_0) = BG_1$ , one can always find anneal conditions  $A_1$  such that  $A_1 \rightarrow A_2(BG_0) = BG_2$ . This follows from the assumption that the sequence  $A_1 \rightarrow A_2$  can be replaced  
5 with a single anneal under different conditions producing an equivalent shift, as stated above. Therefore, in a general case of  $N$  bandgaps, to define the  $n$ th bandgap, one can represent the required shift  $BG_n$  as the superposition of anneal  $A_{N-n+1}$  and the combination of all the subsequent anneals  $A_{N-n+2} \rightarrow \dots \rightarrow A_N$ , all of which, again, can be replaced with a single-stage  
10 anneal. This case is thus reduced to that of  $N=2$ , where the feasibility of the process has been proven.

In order to realise the multiple anneal process with specific bandgap targets in mind, a set of anneal conditions (anneal temperature and time) must be  
15 established for each anneal stage. The development of conditions for the multiple anneal process can be quite time and labour-consuming. Thus a strategic approach is proposed, designed to obtain the required set of conditions with minimum effort.

20 Preferably, the multiple anneal process development is carried out using small test pieces, which are cleaved from a larger piece of unannealed material having a blanket QWI-initiating cap. This way, no multiple photolithography / sputtering steps are required. The method is illustrated in the block diagram of figure 7.

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In step 71, anneal conditions for step  $A_N$  are established in order to obtain the least-shifted bandgap  $BG_1$ .

In step 72, the resulting samples intermixed to  $BG_1$  are used to establish the conditions for step  $A_{N-1}$ . Note the reverse anneal order:  $A_N \rightarrow A_{N-1}$  as opposed to  $A_{N-1} \rightarrow A_N$ .

- 5 In step 73, the anneals are then performed in the correct order:  $A_{N-1} \rightarrow A_N$  using as-grown samples  $BG_0$ . The two shifts obtained by annealing in reverse and normal order are then compared and process symmetry is ascertained. The process is symmetric, i.e. commutative, if the order in which anneals are performed has no effect on the aggregate shift. This
- 10 largely depends on the combination of the material and the core dual-bandgap process being used. The process is described as asymmetric, i.e. non-commutative, if the order in which anneals are performed affects the total shift.
- 15 In step 74, if the process is symmetric, the anneal conditions for the next step  $A_{N-2}$  can be established using the  $BG_2$  bandgap samples obtained at the previous stage. This greatly simplifies the tuning procedure to find the anneal conditions required, as several  $BG_2$  bandgap samples can be prepared simultaneously and then used to search for optimum anneal parameters  $A_{N-2}$
- 20 as in the simplest single-anneal case.

In step 75, if further bandgaps are required, the procedure of step 74 is performed repetitively for each subsequent bandgap.

- 25 In step 76, the conditions for all steps are verified by performing all anneals in the correct order:  $A_1 \rightarrow A_2 \rightarrow \dots \rightarrow A_N (BG_0) = BG_N$ .

- In step 77, if the conditions for the tests in steps 73 and / or 76 are not satisfied, then the process is non-symmetric, and a rigorous tuning procedure
- 30 must be adopted. The procedure then only permits use of virgin  $BG_0$

samples for the development of each step. Specifically for bandgap  $BG_2$ , each iteration of parameters  $A_{N-1}$  must be followed by the previously established anneal  $A_N$ . Only then can the PL be measured and  $A_{N-1}$  adjusted.

- 5 In step 78, if further bandgaps are required, the procedure of step 77 is performed repetitively for each subsequent bandgap. In a general case, for bandgap  $BG_{n+1}$ , each iteration of corresponding anneal parameters  $A_{N-n}(BG_0)$  must be followed by all of the previously established anneals  $A_{N-m} \rightarrow \dots \rightarrow A_N$  (where  $m = (n-1) \dots 1$ ) before any adjustments to  $A_{N-n}$  can be  
10 made.

In step 76, if the above procedure is performed correctly, then the verification condition  $A_1 \rightarrow A_2 \rightarrow \dots \rightarrow A_N(BG_0) = BG_N$  should be satisfied.

- 15 In step 79, the complete multiple anneal process should be verified on a full wafer using the flow of figure 3 and the established anneal parameters  $A_1 \dots A_N$ .

- It has been established that in the majority of multiple anneal processes, the  
20 process is found to be non-symmetric, thus necessitating adherence to the rigorous procedure of steps 77 and 78.

- It is preferred that the duration of all anneal steps  $A_1 \dots A_N$  should be equal, while any QWI shift adjustments are made by varying the temperature only.  
25 This way, the risk of the process performing differently on a full wafer compared to small test pieces is minimised, as for short anneal times the intermixing rate may be affected by the sample size.

- Although the exemplary embodiments above have been described in  
30 connection with an impurity-based QWI-initiating layer, other types of

QWI-initiating materials and techniques may be used to initiate, accelerate or promote the intermixing process. These include impurity-free dielectric caps, sputtered materials (e.g. silica), plasma/sputter damage and some of the techniques outlined in the review of the prior art above.

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Although the exemplary embodiments above have been described in connection with a QWI-inhibiting layer of PECVD silica layer, other types of QWI-inhibiting materials may be used to inhibit, suppress or otherwise retard the intermixing process. These include spin-on glass, sputtered silica etc.

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Embodiments of the invention offer a number of advantages. No QWI-initiating later stripping is required after each intermixing step. They can all be stripped after the final step. No QWI-inhibiting layer deposition step is required after each intermixing step. The initially deposited layer can be relied upon. It is possible to create a large number of bandgap shifts of arbitrary magnitude and these can readily be altered. The process can be made compatible with a wide range of different materials systems including Al-quaternary InP material systems on a semi-insulating InP substrate.

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Other embodiments are intentionally within the scope of the accompanying claims.

## Annex 1: References

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